

Synthesis of Monophasic SrAl₄O₇ Compounds Doped with Europium and Dysprosium and Investigation of Luminescent Properties

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Introduction

The crystal structure of strontium aluminate SrAl₄O₇ is monoclinic with space group C2/c (No. 15) and having cell parameters $a = 13.04$, $b = 9.01$, $c = 5.55$ and $\beta = 106.502^\circ$ [1]. It also has a high-pressure form called β -SrAl₄O₇. Compared to other strontium aluminates (SrAl₂O₄, Sr₃Al₂O₆, SrAl₁₂O₁₉ and Sr₄Al₁₄O₂₅), it is less described in scientific literature. Capron and Douy suggested the synthesis of SrAl₄O₇ from a spray-dried amorphous precursor and specified its stability domains [2]. The results showed that kinetics of formation of this phase is very low, consequently the crystallization of SrAl₄O₇ is sensitive to the heating rate [2].

Luminescence of europium differ significantly depending on its species. Emission of Eu³⁺ is due to $f-f$ transitions, thus emission spectra consist of sharp lines and usually such phosphors emits red light. On the other hand, emission of Eu²⁺ is determined by $d-f$ transitions and emission spectra exhibit wide bands. This type of electron transitions are dependent on crystal field, so color of emission might vary from UV to red light. Furthermore, Eu²⁺ ions often exhibit persistent luminescence phenomenon and such materials are leaders in terms of afterglow times.

Experimental

Strontium aluminate samples were prepared by the conventional solid state reaction. Al₂O₃, SrCO₃, H₃BO₃, Eu₂O₃, and Dy₂O₃ were used as starting materials. The stoichiometric amounts of reagents were thoroughly ground in an agate mortar and then sintered in a furnace at 950 °C for 4 h in reducing atmosphere (*formiergas*). Obtained powders were thoroughly ground in an agate mortar and used for characterization.

Powder X-ray diffraction (XRD) analysis has been carried out employing a Rigaku MiniFlexII diffractometer using $\theta/2\theta$ geometry and CuK α radiation. All of the samples were phase pure, thus XRD diffractograms are not shown in this poster.

Photoluminescence (PL) characterizations were performed on Edinburgh Instruments FLS980 spectrometer equipped with double excitation and emission monochromators, 450 W Xe arc lamp, a cooled (−20 °C) single-photon counting photomultiplier (Hamamatsu R928) and mirror optics for powder samples. The excitation spectra were corrected by a reference detector. In both cases step width was 0.5 nm and integration time was 0.4 s. Afterglow decay measurements were carried out on the same equipment. The samples were excited with 370 nm wavelength light for 60 s, then shutter was shut-off and emission intensities at 475 nm were measured.

References

- [1] A.J.Lindop, D.W.Goodwin. Acta Crystallogr., Sect. B: Struct. Crystallogr. Cryst. Chem. 1972, 8, 2625–2626.
- [2] M.L.Capron, A.Douy. J. Am. Ceram. Soc. 2002, 12, 3036–3040.

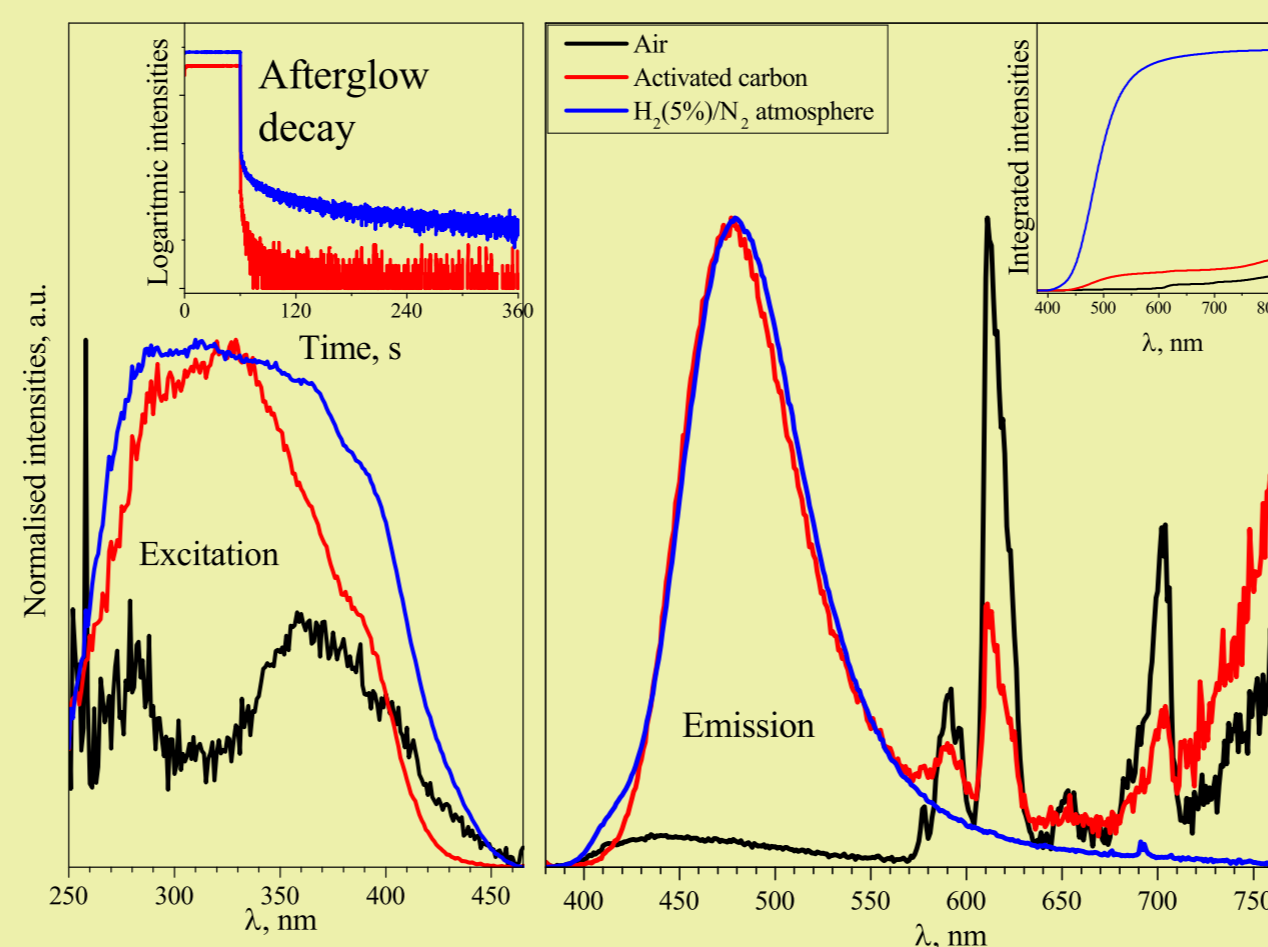
Acknowledgements

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Reduction of Eu³⁺ to Eu²⁺

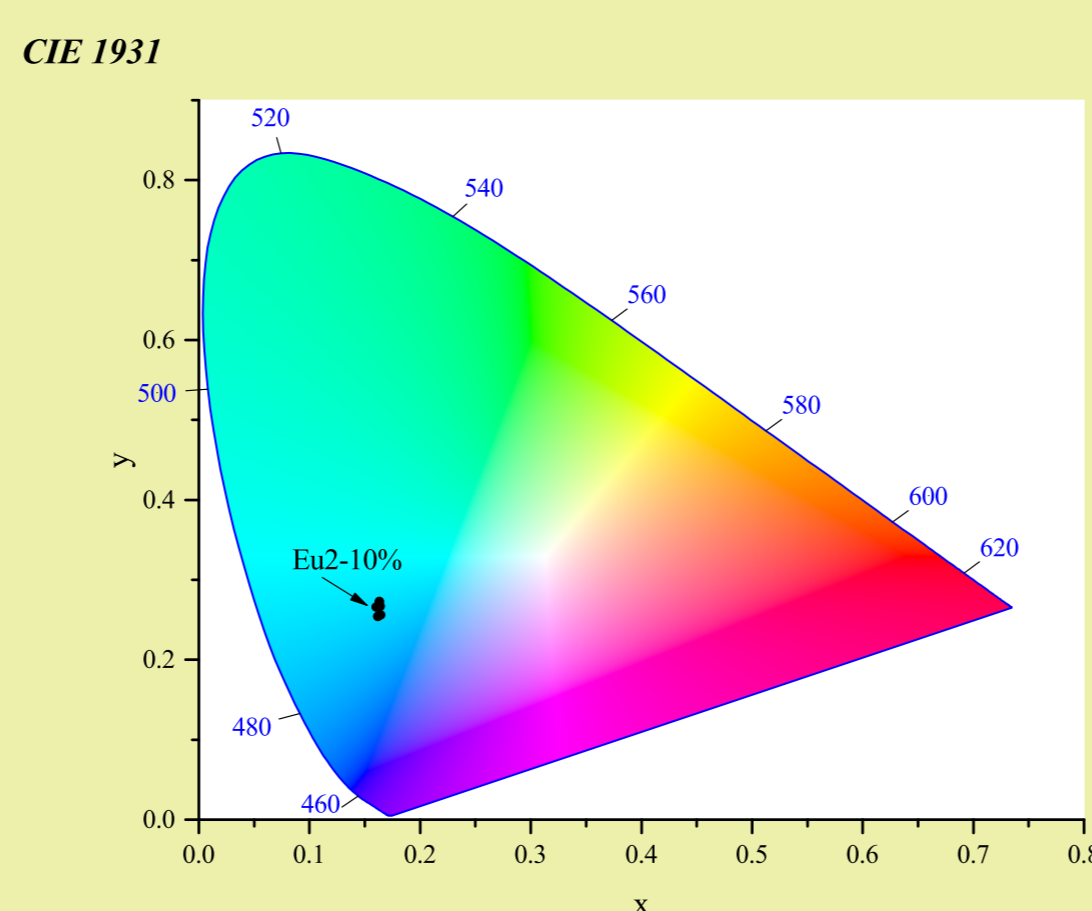
In order to reduce europium from Eu³⁺ to Eu²⁺, three different samples were prepared – one by calcining simply in air atmosphere, second – in reducing atmosphere achieved by burning of activated carbon, and last – by using tube furnace and calcining in *formiergas* (H₂(5 %)/N₂) atmosphere. Results of PL measurements are presented in figures below. As one can see, PL of sample heated in air is typical to luminescence of Eu³⁺ and the sample obtained in H₂/N₂ atmosphere show only one wide emission band (λ_{max} at 475 nm). The PL of sample prepared by heating with activated carbon show both – wide emission band and sharp lines (although of lower intensities), thus it contains a mixture of Eu³⁺ and Eu²⁺ ions. Moreover, stronger reducing atmosphere significantly increased afterglow decay times.



Excitation ($\lambda_{em} = 475$ nm) and emission ($\lambda_{ex} = 370$ nm) spectra of Sr_{0.98}Al₄O₇:Eu_{0.02} samples prepared by annealing at different atmospheres. Right inset: integrated intensities of emission spectra. Left inset: afterglow decay curves.

Colors

Color coordinates were calculated from emission spectra and plotted on CIE1931 diagram. All of the samples fall within greenish blue region.



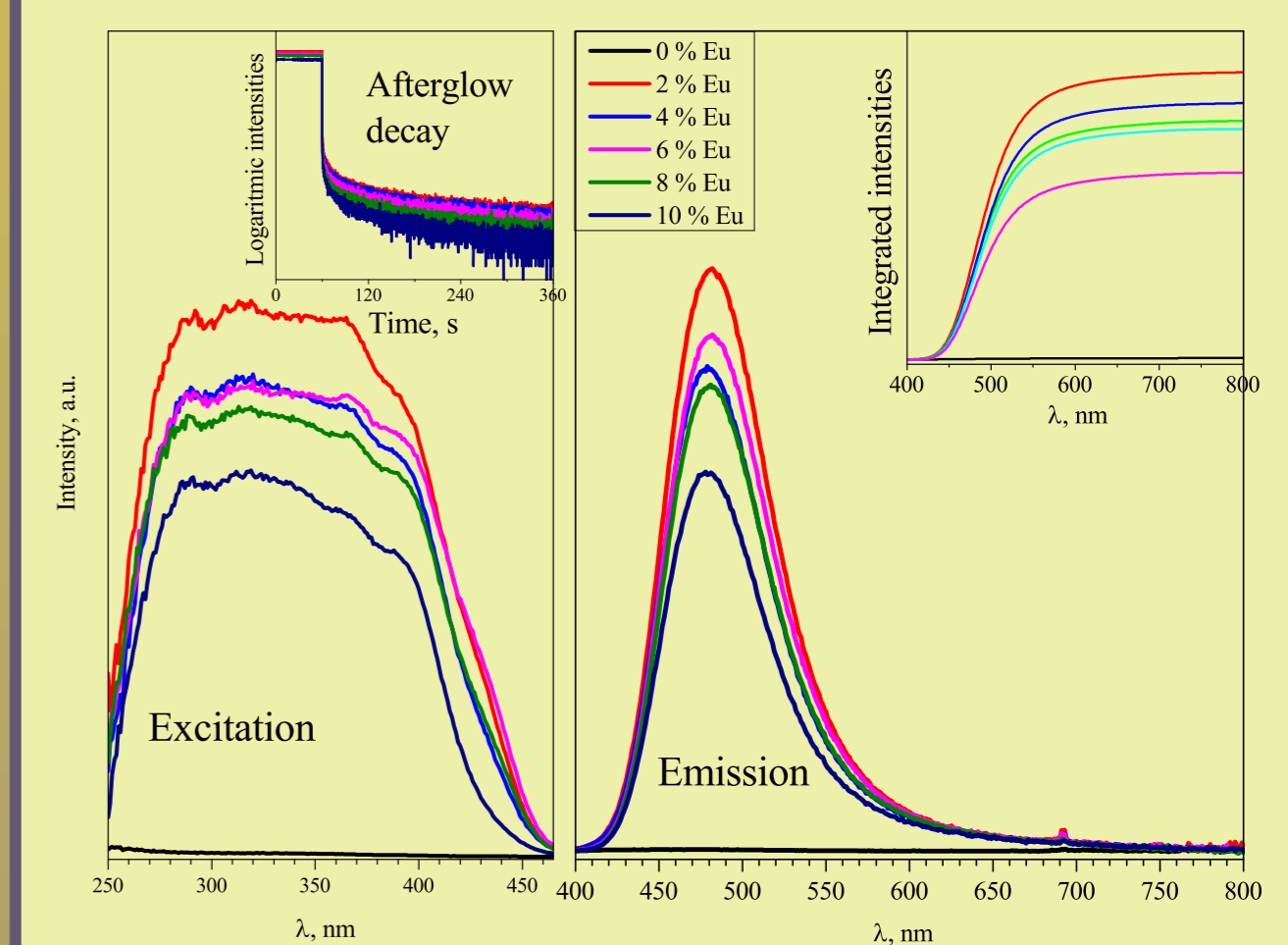
CIE 1931 chromaticity diagram of the Sr_{1-x}Al₄O₇:Eu_x samples.

Conclusions

Solid state synthesis is a suitable way for the preparation of monophasic SrAl₄O₇ doped with Eu and co-doped with Dy. Reducing atmosphere achieved by using 5 % H₂/N₂ gas mixture is sufficient for the reduction of Eu³⁺ to Eu²⁺. Greenish blue emission maxima of europium doped samples is at 475 nm and 2 % doped sample exhibit highest intensity and persistent luminescence with afterglow lasting up to 5 minutes. Co-doping with Dy³⁺ significantly increase afterglow duration up to 30 min. Moreover, high concentrations of dysprosium shifts emission maxima to 512 nm.

Influence of Eu²⁺ concentration

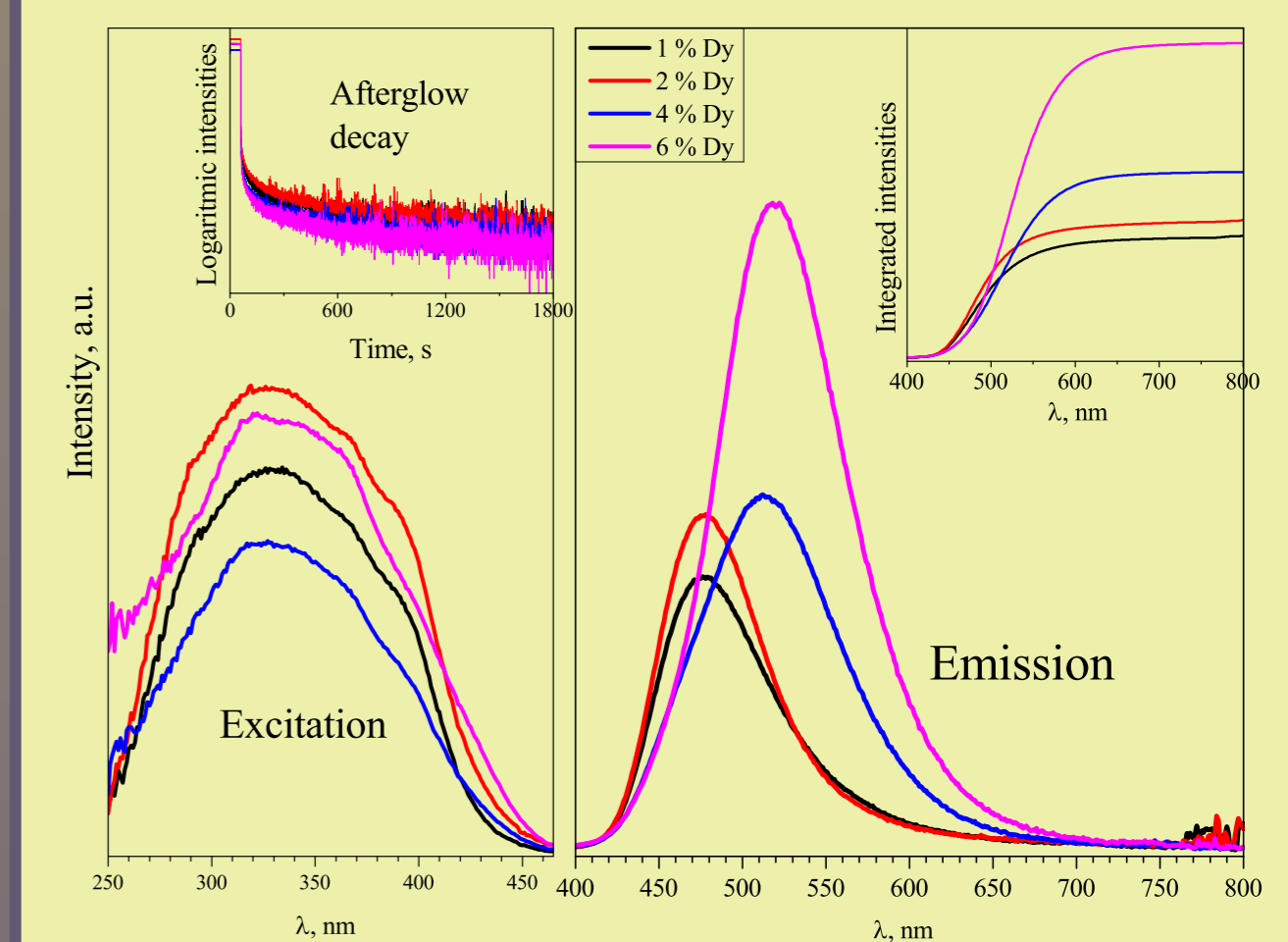
All of the samples (except undoped one) exhibit broad emission peaking at 475 nm. The sample doped with 2 % Eu²⁺ exhibited highest intensities and strongest persistent luminescence.



Excitation ($\lambda_{em} = 475$ nm) and emission ($\lambda_{ex} = 370$ nm) spectra of Sr_{1-x}Al₄O₇:Eu_x samples. Right inset: integrated intensities of emission spectra. Left inset: afterglow decay curves.

Co-doping with Dy³⁺

Different concentrations of dysprosium was co-doped in Sr_{0.98}Al₄O₇:Eu_{0.02} samples. Interestingly, higher Dy³⁺ concentrations shifted emission maxima towards higher wavelength. Highest co-dopant concentration yielded highest PL intensities, while strongest persistent luminescence was observed in 2 % Dy sample.



Excitation ($\lambda_{em} = 475$ nm) and emission ($\lambda_{ex} = 370$ nm) spectra of Sr_{0.98-x}Al₄O₇:Eu_{0.02},Dy_x samples. Right inset: integrated intensities of emission spectra. Left inset: afterglow decay curves.